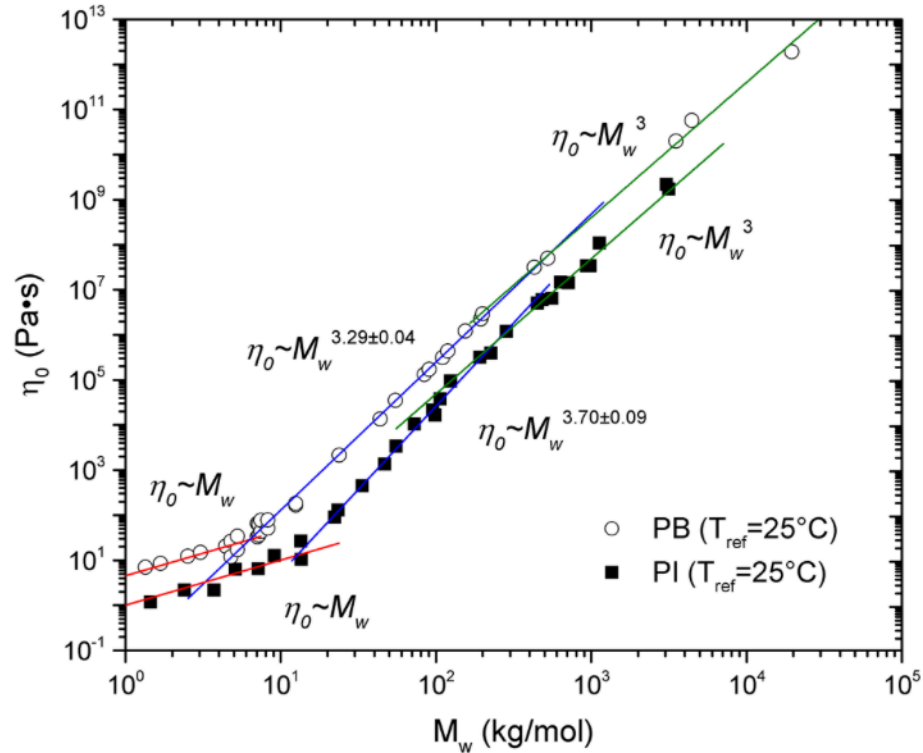


1) Unidat, Richter, and Fetters [*Macro.* **48** 6638 (2015)] draw connections between the packing length,  $p$ , and certain rheological transitions they have found in polymer melts. Unidat describes three regimes seen in Figure 1 below: 1.) Rouse behavior for  $M < M_c$ ; 2.) Reptation with constraint release and contour fluctuations between  $M_c$  and  $M_r$ ; and 3.) Pure reptation behavior for high molecular weights  $M > M_r$ .



**Figure 1.** Experimental data for the zero-shear viscosity as a function of molecular weight of polybutadiene (PB) and polyisoprene (PI) melts. Data are obtained from various sources<sup>9,11,12</sup> and show the three scaling regimes. Lines show the different power law fits for each regime.

- Give an equation that defines the packing length.
- The entanglement molecular weight,  $M_e$ , is determined by the plateau modulus,  $G_0 = \rho RT/M_e$ . Derive this equation from the Gaussian probability function and explain why  $M_e$  is related to the plateau modulus.
- Unidat relates  $M_c$  and  $M_r$  from Figure 1 with  $M_e$  using,  $M_c = M_e[p^*/p]^{0.65}$  and  $M_r = M_e[p^*/p]^{3.9}$ , where  $p$  is the packing length, and  $p^*$  is a constant of about 12 Å. What happens to the behavior seen in Figure 1 as  $p$  approaches  $p^*$ ? Can  $p$  be larger than  $p^*$ ? Explain.
- Unidat describes the tube diameter,  $a$ , as “the end-to-end distance of an entanglement strand” citing Doi and Edwards. Explain this definition of the tube diameter. What happens to the tube at long times,  $t > \tau_{\text{entanglement}}$ . How is  $\tau_{\text{entanglement}}$  affected by  $M_w$ ?
- Three parameters on the order of 1 nm are used to describe local polymer structure, the tube diameter,  $a$ , the packing length,  $p$ , and the Kuhn length,  $l_K$ . Explain what the Kuhn length is and how it relates to the persistence length,  $l_p$ .

2) The figure below by Wilding, Müller, and Binder shows a phase diagram for a polymer solution.

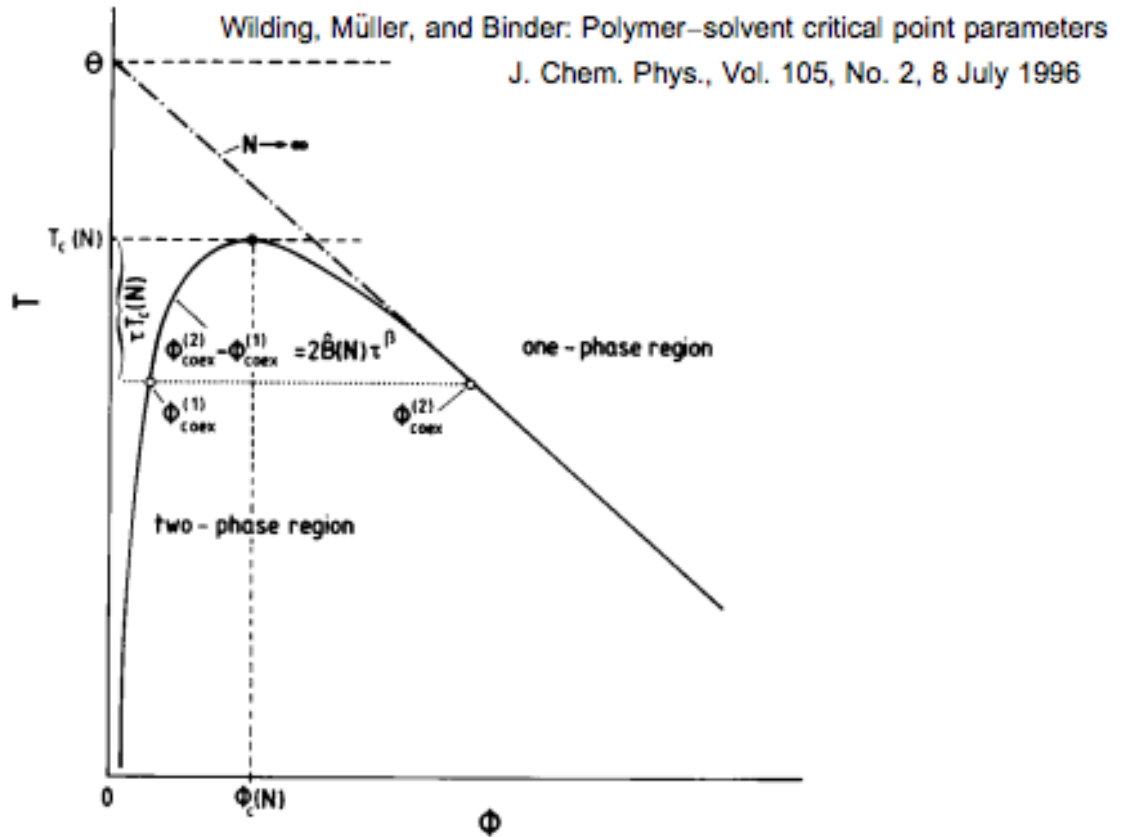


FIG. 1. Schematic phase diagram of a polymer solution in the space of the temperature  $T$  and the volume fraction  $\phi$ . The coexistence curve separates a dilute solution of collapsed chains [at  $\phi_{\text{coex}}^{(1)}$ ] from a semidilute solution of overlapping chains [at  $\phi_{\text{coex}}^{(2)}$ ]. These two branches of the coexistence curve merge at a critical point  $T_c(N)$ ,  $\phi_c(N)$ . For  $N \rightarrow \infty$  the critical point merges with the  $\Theta$  point of a dilute polymer solution [ $T_c(N \rightarrow \infty) \rightarrow \Theta$ ,  $\phi_c(N \rightarrow \infty) \rightarrow 0$ ] and the unmixing transition has a tricritical character. At  $T = \Theta$ , the chain configurations are ideal Gaussian coils, while their structure at  $T_c(N)$  is nontrivial.

- Give an equation that describes the behavior of Figure 1.
- How are the critical temperature and critical composition determined from the equation of part a)?
- Give an equation for the overlap concentration,  $\phi^*$ . How would  $\phi^*$  impact the phase diagram shown in Figure 1?
- Explain what the caption means when it mentions that the chain in the collapsed state is Gaussian while in solution it is an expanded coil. Why is the structure “non-trivial” at the critical point?
- Is it possible to have a thermodynamically stable *dilute* polymer solution below the theta temperature? Why does the theta temperature “have a tricritical character”?